

Unified picture of the doping dependence of superconducting transition temperatures in alkali metal/ammonia intercalated FeSe

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In the recently synthesized $\text{Li}_x(\text{NH}_2)_y(\text{NH}_3)_z\text{Fe}_2\text{Se}_2$ family of iron chalcogenides, a molecular spacer consisting of lithium ions, lithium amide, and ammonia separates the layers of FeSe. It has been shown that upon variation of the chemical composition of the spacer layer, superconducting transition temperatures can reach $T_c \sim 44$ K, but the relative importance of the layer separation and effective doping to the T_c enhancement is currently unclear. Using state of the art band structure unfolding techniques, we construct eight-orbital models from *ab initio* density functional theory calculations for these materials. Within an RPA spin-fluctuation approach, we show that the electron doping enhances the superconducting pairing, which is of s_{\pm} symmetry and explain the experimentally observed limit to T_c in the molecular spacer intercalated FeSe class of materials.

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After the discovery of iron-based superconductors in 2008, transition temperatures were quickly improved to ~ 56 K by chemical substitution [1]. Recently, the possible discovery of superconductivity with $T_c = 65$ K [2] and even $T_c \sim 100$ K [3] in single-layer FeSe films grown by molecular beam epitaxy on SrTiO₃ showed that temperatures close to and above the boiling point of liquid nitrogen (77 K) might be achievable. These results have initiated an intensive debate regarding the origin of the high superconducting temperatures and the role played by electron doping via substrate, dimensionality, and lattice strain.

While bulk FeSe has a T_c of only 8–10 K, it has been known for some time that it can be substantially enhanced, to 40 K or higher by alkali intercalation [4]. Materials with a single alkali $A = \text{K}, \text{Cs}, \text{Rb}$ between FeSe layers of nominal form $A_x\text{Fe}_{2-y}\text{Se}_2$ have been intensively studied, and shown to display a wide variety of unusual behaviors relative to the Fe pnictide superconducting materials [5]. These include likely phase separation into an insulating phase with block antiferromagnetism and ordered Fe vacancies, and a superconducting phase that is strongly alkali deficient and whose Fermi surface, as measured by ARPES, apparently contains no holelike Fermi surface pockets, in contrast to Fe pnictides. Since the popular spin fluctuation scenario for s_{\pm} pairing relies on near nesting of hole and electron pockets, it has been speculated that a different mechanism for pairing might be present in these materials, and even within the spin fluctuation approach, different gap symmetries including d -wave pairing have been proposed [6–9]. However, the gap symmetry and structure is still controversial [10,11].

In addition to the unusual doping, speculation on the origin of the higher T_c has centered on the intriguing possibility that enhancing the FeSe layer spacing improves the two-dimensionality of the band structure and hence Fermi surface nesting [12,13]. In an effort to investigate the latter effect, organic molecular complexes including alkalis were recently intercalated between the FeSe layers [12–19], yielding transition temperatures up to 46 K. The most intensively

studied materials incorporate molecules including ammonia, for example, $\text{Li}_{0.56}(\text{NH}_2)_{0.53}(\text{NH}_3)_{1.19}\text{Fe}_2\text{Se}_2$ with $T_c = 39$ K [14] and $\text{Li}_{0.6}(\text{NH}_2)_{0.2}(\text{NH}_3)_{0.8}\text{Fe}_2\text{Se}_2$ with $T_c = 44$ K [15]. The crystal structure of a stoichiometric version of these materials is shown in Fig. 1. Recently, Noji *et al.* [13] compared data on a wide variety of FeSe intercalates and noted a strong correlation of T_c with interlayer spacing, corresponding to a nearly linear increase between 5 to 9 Å, followed by a rough independence of spacing with further increase between 9 to 12 Å.

In the present work, we study the question of how exactly doping and interlayer distance influence T_c in molecular intercalates of FeSe, whether these effects are separable, and what gives rise to the apparent upper limit for T_c in this family of iron chalcogenides. Using a combination of first-principles calculations of the electronic structure of several materials in this class, together with model calculations of spin fluctuation pairing based on these band structures, we argue that the strength and wave vector of spin-fluctuations in lithium/ammonia intercalated FeSe can be controlled by tuning the $\text{Li}^+:\text{NH}_2^-$ ratio in the spacer layer. We show that the evolution of T_c with electron doping can be understood from the shape of the density of states close to the Fermi level. As long as hole pockets are present, we find that the superconducting pairing is of s_{\pm} character and identify a subleading $d_{x^2-y^2}$ instability. We believe that our interpretation is valid in a broad class of related materials.

We performed density functional theory calculations for $\text{Li}_{0.5}(\text{NH}_2)_y(\text{NH}_3)_z\text{Fe}_2\text{Se}_2$ at various ratios of NH_2^- to NH_3 content, starting from the experimentally determined structures $\text{Li}_{0.56}(\text{NH}_2)_{0.53}(\text{NH}_3)_{1.19}$ [14] and $\text{Li}_{0.6}(\text{NH}_2)_{0.2}(\text{NH}_3)_{0.8}$ [15], which include fractionally occupied atomic sites for lithium, hydrogen, and nitrogen. In order to accommodate the experimental stoichiometry, we construct a $2 \times 1 \times 1$ (4 Fe) supercell for the former, and a $2 \times 2 \times 1$ (8 Fe) supercell for the latter compound. We replace all fractionally occupied nitrogen positions by fully occupied positions. As hydrogen positions are not known precisely from experiment, we arrange the hydrogen atoms so that we obtain NH_3 groups with angles of about 108° as encountered in ammonia and further relax these positions within the local density approximation (LDA)

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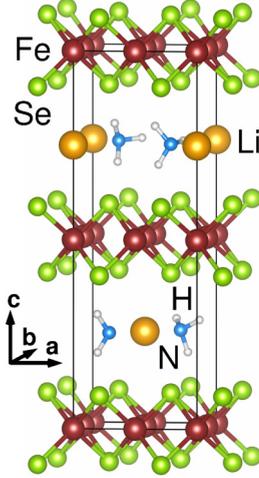


FIG. 1. (Color online) Idealized crystal structure of $\text{Li}_{0.5}(\text{NH}_3)\text{Fe}_2\text{Se}_2$. For a detailed discussion of experimental crystal structures see Refs. [14,15].

[20] with the projector augmented wave (PAW) basis [21] as implemented in GPAW [22] until forces are below 2 meV/Å. In the $2 \times 1 \times 1$ supercell, we place the lithium atom in one half of the unit cell and leave the lithium position in the other half unoccupied. In the $2 \times 2 \times 1$ supercell, we arrange the lithium atoms in a checkerboard pattern of occupied and vacant sites (Fig. 1).

Initially, we only consider charge neutral NH_3 ammonia groups in the spacer and no NH_2^- . In this way, we obtain idealized structures with formula units $\text{Li}_{0.5}(\text{NH}_3)\text{Fe}_2\text{Se}_2$ and $\text{Li}_{0.5}(\text{NH}_3)_2\text{Fe}_2\text{Se}_2$ where chalcogen height and unit cell parameters are chosen as in the experimental structures [14,15]. Both structures belong to the space group P1 because of NH_3 situated in the spacer layer. Note that by setting up both structures with neutral NH_3 , we are able to disentangle possible effects of the structural differences from the effect of doping through the composition of the spacer layer.

The experimentally available samples [14,15] contain both NH_3 and NH_2^- . The radical NH_2^- neutralizes the charge donated to the FeSe layer by Li^+ and reduces the doping level. In order to capture this compensation of charge in our simulations, we use the virtual crystal approximation (VCA) [23] starting from supercells $\text{Li}_{0.5}(\text{NH}_3)\text{Fe}_2\text{Se}_2$ and $\text{Li}_{0.5}(\text{NH}_3)_2\text{Fe}_2\text{Se}_2$, which correspond to the maximally electron doped compounds. The use of VCA has the advantage that doping is treated in a continuous and rather isotropic, but not rigid band fashion. We checked these calculations by removing hydrogen atoms explicitly instead of doing VCA and found the differences to be negligible.

The analysis of the band structure of these supercells is done within an all-electron full-potential local orbital (FPLO) [24] basis and we use LDA as exchange-correlation functional [20]. We then use projective Wannier functions as implemented in FPLO [25] to downfold the band structure. In our tight-binding models, we keep the Fe 3d and Se 4p states. In order to obtain the band structure and Fermi surface of the supercells in the conventional two-iron unit cell, we use our recently developed technique [26] to translationally

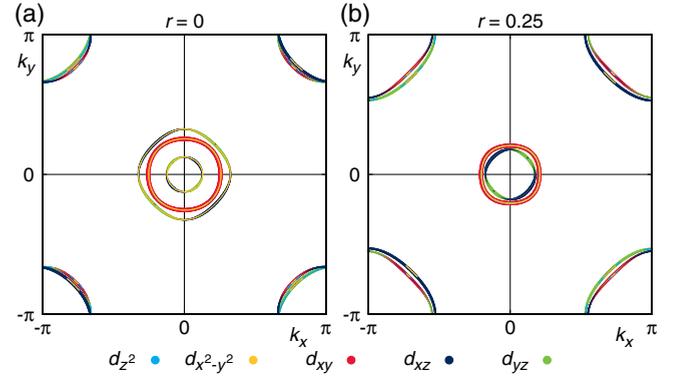


FIG. 2. (Color online) Fermi surface in the 16-band tight-binding model for (a) $r = 0.0$ and (b) 0.25 in the two-Fe Brillouin zone at $k_z = 0$. The colors indicate the weights of Fe 3d states.

unfold the 32 and 64 band supercell models to a 16-band model of the two-Fe equivalent Brillouin zone. For calculations of susceptibility and superconducting pairing, we use subsequent glide reflection unfolding [26] of the bands to obtain the eight-band model of the one-Fe equivalent Brillouin zone.

First, we investigated the properties of the maximally electron doped compounds in our study, $\text{Li}_{0.5}(\text{NH}_3)\text{Fe}_2\text{Se}_2$ (ammonia poor) and $\text{Li}_{0.5}(\text{NH}_3)_2\text{Fe}_2\text{Se}_2$ (ammonia rich). At the Fermi level (not shown), both feature two large electron pockets in the corners of the two-Fe Brillouin zone and two small hole pockets around Γ . This confirms that the lithium atoms donate electrons to the FeSe layer. Both systems have the same electron doping but different interlayer spacing. This is observed in the k_z dispersion of the Fermi surface, where the smaller interlayer distance of the ammonia-poor compound leads to a slightly increased corrugation of the cylinders.

In the experimentally realized compounds $\text{Li}_{0.56}(\text{NH}_2)_{0.53}(\text{NH}_3)_{1.19}$ and $\text{Li}_{0.6}(\text{NH}_2)_{0.2}(\text{NH}_3)_{0.8}$, the spacer layer nominally donates a charge of 0.015 and 0.2

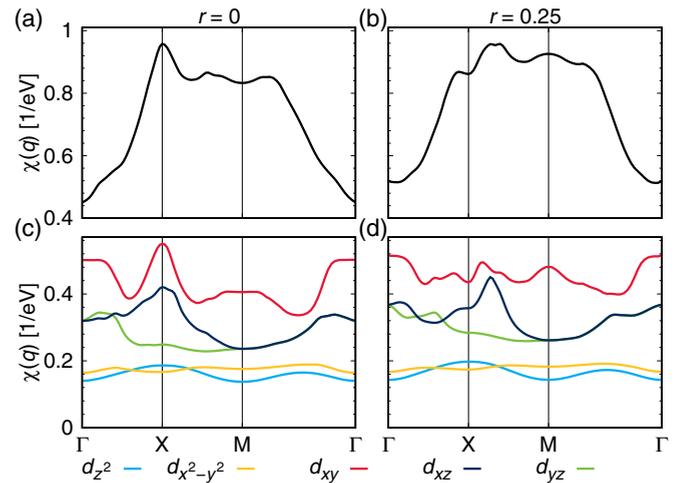


FIG. 3. (Color online) Summed static susceptibility (top) and its diagonal components χ^{aa} (bottom) in the eight-band tight-binding model for [(a) and (c)] $r = 0.0$ and [(b) and (d)] 0.25 in the one-Fe Brillouin zone. The colors identify the Fe 3d states.

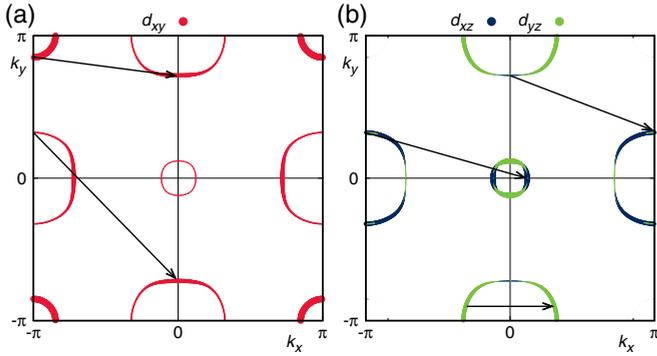


FIG. 4. (Color online) Fermi surface in the eight-band tight-binding model for $r = 0.25$ in the one-Fe Brillouin zone at $k_z = 0$. Shown are the orbital characters for the d_{xy} (a) and the $d_{xz/yz}$ (b) orbitals. The arrows represent dominant interaction vectors identified from peaks in the susceptibility.

electrons per iron atom, respectively. These doping levels are lower than in our model materials $\text{Li}_{0.5}(\text{NH}_3)\text{Fe}_2\text{Se}_2$ and $\text{Li}_{0.5}(\text{NH}_3)_2\text{Fe}_2\text{Se}_2$. To investigate the doping dependence of the electronic structure at a given interlayer spacing, we consider $\text{Li}_{0.5}(\text{NH}_3)\text{Fe}_2\text{Se}_2$ and hole dope it by means of the virtual crystal approximation as explained above. To simplify the notation, we label compounds from now on not by their full chemical formula, but by an index $r = \{0.0, \dots, 0.25\}$, which refers to the chemical formula $\text{Li}_{0.5}(\text{NH}_2)_{0.5-2r}(\text{NH}_3)_{0.5+2r}\text{Fe}_2\text{Se}_2$. $r = 0.25$ refers to the compound $\text{Li}_{0.5}(\text{NH}_3)\text{Fe}_2\text{Se}_2$ with maximal electron doping, where lithium nominally transfers a quarter of an electron to each iron atom. Increasing the NH_2^- content immediately brings up a third hole pocket to the Fermi level, which is three-dimensional at intermediate doping and becomes two-dimensional once the charge introduced by lithium is fully compensated by NH_2^- groups. $r = 0$ denotes the compound where the charge introduced by lithium is nominally compensated by NH_2^- and no electrons are donated to the FeSe layer. The Fermi surfaces of the end members ($r = 0.0$ and $r = 0.25$) are shown in Fig. 2. The band structure on high-symmetry paths is included in Ref. [23].

Upon further analysis of the tight-binding parameters, we find that the hole pockets do not only shrink because the electron doping raises the Fermi level, but also because the nearest-neighbor hopping in the Fe $3d_{xy}$ orbital decreases steeply as a function of electron doping. This near cancellation of direct and indirect hopping paths has been discussed in the literature for other iron based superconductors [27–29]. In the materials investigated here, we find that the degree of localization in the Fe $3d_{xy}$ orbital can be tuned with relatively low electron doping. Further information is given in Ref. [23].

Next, we investigate the doping dependence of spin fluctuations. The noninteracting static susceptibility on the high-symmetry path calculated in the one-Fe Brillouin zone for $r = 0.0$ and 0.25 is shown in Fig. 3. The susceptibility χ_{st}^{pq} is a four-tensor in orbital indices. The observable static susceptibility is defined as the sum over all components χ_{aa}^{bb} . In the undoped compound ($r = 0.0$), the structure of the static susceptibility resembles strongly what is found for materials like LaFeAsO or BaFe_2As_2 . The electron doping notably shifts the maximum from $X = (\pi, 0)$ towards $M = (\pi, \pi)$ and the former valley at M transforms into a peak. The absence of a $(\pi, 0)$ peak in electron doped compounds suggests why no orthorhombic phase or stripelike magnetism have been found in FeSe intercalates so far [30].

The shifts of the dominant vectors of spin fluctuations can be understood from the nesting properties and orbital character on the Fermi surface in the one-Fe Brillouin zone. The undoped compound ($r = 0.0$) is dominated by $(\pi, 0)$ nesting of electron and hole pockets, whereas the electron doped compound ($r = 0.25$) (see Fig. 4) features scattering between electron and hole pockets with altered wave vector competing with scattering between electron pockets. The dominant contributions to the static susceptibility originate from the d_{xy} and $d_{xz/yz}$ orbitals.

To explore how the superconducting state might depend on interlayer spacing and doping, we use the 3D version of random phase approximation (RPA) spin fluctuation theory [31] with Hamiltonian $H = H_0 + H_{\text{int}}$. Here H_0 is the tight-binding Hamiltonian derived from the DFT calculations using the projective Wannier function formalism described above,

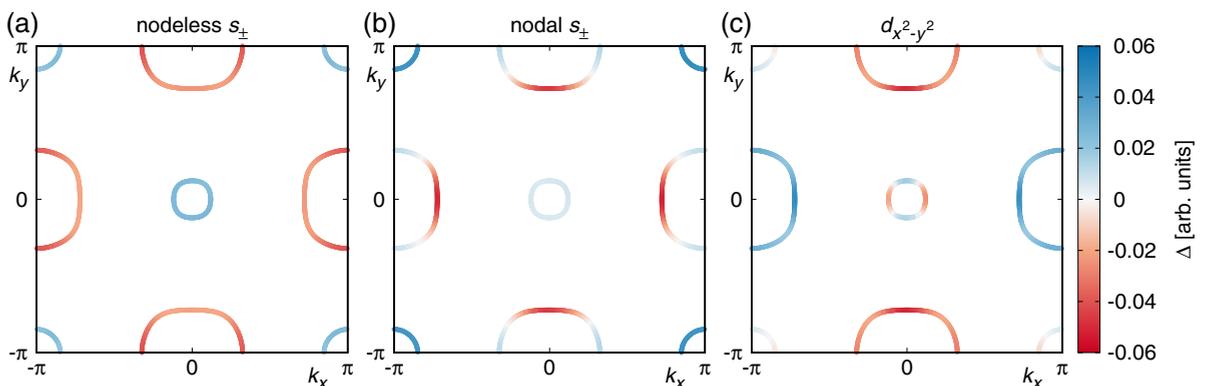


FIG. 5. (Color online) Solutions of the linearized gap equation on the Fermi surface within the 8-band tight-binding model for $r = 0.25$ in the one-Fe Brillouin zone at $k_z = 0$. The relevant instabilities are (a) nodeless s_{\pm} , (b) nodal s_{\pm} , and (c) $d_{x^2-y^2}$. We assume spin rotation-invariant interaction parameters $U = 1.35$ eV, $U' = U/2$, and $J = J' = U/4$.

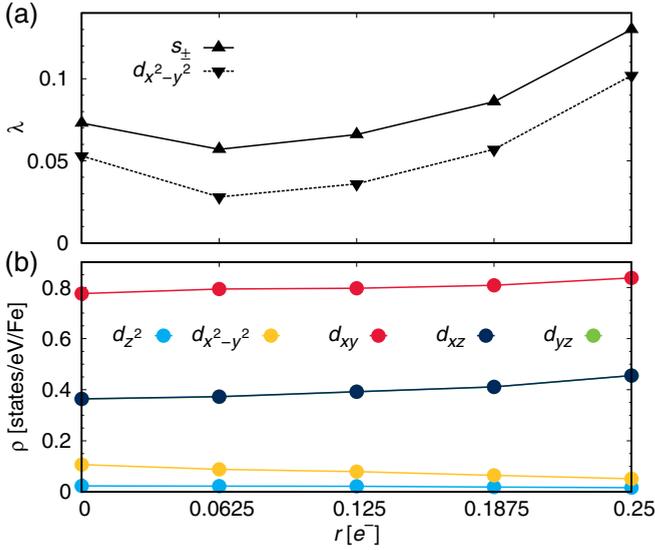


FIG. 6. (Color online) Trend of the eigenvalues of s_{\pm} and $d_{x^2-y^2}$ solutions (a) and orbital resolved Fe 3d density of states at the Fermi level (b) with doping.

and H_{int} is the Hubbard-Hund interaction, including the onsite intra (inter) orbital Coulomb interaction U (U'), the Hund's rule coupling J and the pair hopping energy J' . We keep the selenium states in the entire calculation, but consider interactions only between Fe 3d states. We assume spin rotation-invariant interaction parameters $U = 1.35$ eV, $U' = U/2$, and $J = J' = U/4$. The effective interaction in the singlet pairing channel is then constructed via the multiorbital RPA procedure. Both the original Hamiltonian and effective interaction are discussed, e.g., in Ref. [32].

For all values of electron doping (structures $r = 0.0$ to 0.25) and interlayer spacing [structures $\text{Li}_{0.5}(\text{NH}_3)\text{Fe}_2\text{Se}_2$ and $\text{Li}_{0.5}(\text{NH}_3)_2\text{Fe}_2\text{Se}_2$] considered, we find the leading instability to be of nodeless s_{\pm} character, while subleading solutions are of nodal s_{\pm} and $d_{x^2-y^2}$ type (see Fig. 5 for structure $r = 0.25$). These are the leading states expected in the case of a nearly 2D system with both hole and electron pockets. Repulsive electron-hole $d_{xz/yz}$ and d_{xy} interactions favor nodeless s_{\pm} pairing, while interelectron pocket interactions, orbital weight variations around the Fermi surface, and intraband Coulomb interactions are known to frustrate the s_{\pm} interaction and drive nodal behavior and eventually d -wave interactions when hole pockets disappear [8,33].

We observe that the source of the moderate quantitative enhancement of T_c with electron doping lies in an increased density of states at the Fermi level. For both the d_{xy} and the $d_{xz/yz}$ orbitals, the slope of density of states near the Fermi level is positive [Fig. 6(b)] so that electron doping leads to an enhanced susceptibility and superconducting pairing strength

as the doping approaches the edge of the hole bands, which appears as a sharp drop of the d_{xy} DOS [23]. The small initial decrease of the pairing eigenvalue at low electron doping [Fig. 6(a)] is a consequence of the degraded nesting.

Alternatively, when we keep the electron doping levels fixed to the same value and analyze only the interlayer spacing effect (structures $\text{Li}_{0.5}(\text{NH}_3)\text{Fe}_2\text{Se}_2$ with $c = 8.1$ Å and $\text{Li}_{0.5}(\text{NH}_3)_2\text{Fe}_2\text{Se}_2$ with $c = 10.3$ Å), we find that the Fermi surface turns completely two-dimensional for a c -axis length between 8.1 and 10.3 Å, where T_c saturates in experiment. Analyzing the susceptibility and superconducting pairing for both structures, we find no qualitative differences. Quantitatively, the perfectly two-dimensional Fermi surface of the ammonia rich compound leads to an increased susceptibility and larger pairing eigenvalue than in the ammonia poor compound. The increased pairing eigenvalue would correspond to an enhanced T_c .

Our calculations show that both increasing electron doping and lattice spacing contribute to enhancing T_c . However, experimentally it is found that the ammonia poor compound (larger electron doping) with smaller c axis shows a higher T_c [$\text{Li}_{0.6}(\text{NH}_2)_{0.2}(\text{NH}_3)_{0.8}\text{Fe}_2\text{Se}_2$, $T_c = 44$ K] than the ammonia rich compound (smaller electron doping) with larger c axis [$\text{Li}_{0.56}(\text{NH}_2)_{0.53}(\text{NH}_3)_{1.19}\text{Fe}_2\text{Se}_2$, $T_c = 39$ K]. Therefore the variations in lattice parameters observed experimentally cannot be the source of the enhancement of T_c . Within our picture, this leaves only the electron doping level as the controlling parameter. Hence we conclude that T_c is mainly controlled by the electron doping level when the Fermi surface is mostly two dimensional. Therefore it is unlikely that T_c can be enhanced further by intercalation of larger molecules.

Summarizing, we investigated the $\text{Li}_x(\text{NH}_2)_y(\text{NH}_3)_z\text{Fe}_2\text{Se}_2$ family of FeSe intercalates and found that the FeSe layer is moderately electron doped. The electron doping moves the Fermi level towards the edge of the hole bands, which gives rise to increased superconducting transition temperatures due to an increase in the density of states at the Fermi level. We also showed that recently achieved interlayer distances in FeSe intercalates already produce a two-dimensional Fermi surface, which is optimal for T_c . Further experimental work should therefore concentrate on the charge doping through the spacer layer.

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